

The Crystal Structure of $(\text{Fe}_x\text{Ni}_{1-x})_{22}\text{Se}_{16}$

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The crystal structure of the compound $(\text{Fe}_x\text{Ni}_{1-x})_{22}\text{Se}_{16}$, ($x=0.2$) has been determined using X-ray methods. The crystal system is body centred tetragonal, with the lattice constants $a=7.208$ and $c=11.393$ Å. The most probable space group is $I4m2$, and the selenium and metal atoms are distributed as follows:

8 Me at (g):	0.275,	0.275,	0
8 Me at (i):	0.278,	0	, 0.326
4 Me at (f):	0	, 0.5	, 0.024
2 Me at (c):	0	, 0.5	, 0.25
8 Se at (i):	0.242,	0	, 0.115
8 Se at (i):	0	, 0.221,	0.349

It was not possible to determine the mutual distribution of iron and nickel atoms.

The existence of the two isomorphous phases $(\text{Co}_x\text{Ni}_{1-x})_{22}\text{Se}_{16}$ and $(\text{Fe}_x\text{Ni}_{1-x})_{22}\text{Se}_{16}$ was reported by Haraldsen *et al.*¹ In a later paper by Røst and Haugsten,² the value of x of the latter phase was given as $0.04 < x < 0.23$ at 580°C . The crystal system of this phase was found to be body centred tetragonal, with the lattice dimensions $a=7.164$ and $c=11.403$ Å at the nickel-rich phase limit, whereas the corresponding values for the iron-rich sample are $a=7.217$ and $c=11.410$ Å. On further increase in the Fe/Ni ratio, another phase region with tetragonal structure appears. The lattice constants of this phase are $a=3.73$ and $c=5.76$ Å at 600°C . The structure is assumed to be of the rickardite ($\text{Cu}_{2.8}\text{Te}_2$) type, which is an intermediate between the $C38(\text{Ni}_2\text{In})$ and the $B10(\text{PbO})$ type structures.

The existence of the two ternary phases $(\text{Fe},\text{Ni})_{22}\text{Se}_{16}$ and $(\text{Co},\text{Ni})_{22}\text{Se}_{16}$ has been confirmed by Stevels *et al.*³ Under special conditions, they also obtained a similar metastable phase in the binary system Ni—Se.

EXPERIMENTAL

Single crystals of $(\text{Fe}_x\text{Ni}_{1-x})_{22}\text{Se}_{16}$ were obtained by the transport reaction method, using iodine as transporting agent. Samples used in an earlier investigation,² with the composition corresponding to $x=0.2$, $(\text{Fe}_{.116}\text{Ni}_{.463}\text{Se}_{.421})$ and traces of iodine were

enclosed in evacuated silica tubes and annealed at $\sim 600^\circ\text{C}$ in a furnace in which there was a small temperature gradient. A number of small crystals were obtained after six weeks treatment, the one employed in the X-ray investigation being ~ 0.05 mm long by 0.02 mm.

Single crystal photographs were obtained using $\text{MoK}\alpha$ -radiation in an integrating Weissenberg camera of 57.3 mm diameter. The multiple film technique was applied, using Sn foils between the films, and intensity data were obtained during rotation of the crystal about the b -axis (the needle axis of the crystal). The determination of structure was based on reflections from the layers $h0l$ to $h6l$, and of a total number of 370 reflections, only 191 were observable. The unobserved reflections were omitted from the final least squares refinements.

The atomic form factors employed were those given by Hanson *et al.*⁴ Those for the metal atoms, were interpolated assuming 20 % Fe and 80 % Ni in each metal position. No corrections were made for absorption or secondary extinction.

CRYSTAL DATA *

$\text{Fe}_{116}\text{Ni}_{463}\text{Se}_{421}$.

Tetragonal

$a = 7.208 \pm 0.002 \text{ \AA}$, $c = 11.393 \pm 0.003 \text{ \AA}$.

Unit cell volume 585.9 \AA^3 .

Observed density 7.135 g cm^{-3} .

Unit cell content 38.01 (38) atoms, *i.e.* 16 Se and 22 Me atoms (average 4.4 Fe and 17.6 Ni atoms).

Systematically absent reflections:

hkl when $h+k+l=2n+1$.

Intensities of hkl and $\bar{h}kl$ were estimated to be equal.

Possible space groups:

$I422$, $I4mm$, $I4m2$, $I42m$, and $I4mmm$.

RESULTS AND DISCUSSION

The iron-rich samples of $(\text{Fe}_x\text{Ni}_{1-x})_{22}\text{Se}_{16}$, and the nickel-rich rickardite-type phase of the Fe—Ni—Se system differ only slightly in gross composition.² The lattice constants of the former phase are approximately double those of the latter, and furthermore, the X-ray powder patterns of the two phases show similarities, even though the larger unit cell of $(\text{Fe}_x\text{Ni}_{1-x})_{22}\text{Se}_{16}$ gives rise to a number of additional reflections. The choice of space group for $(\text{Fe}_x\text{Ni}_{1-x})_{22}\text{Se}_{16}$ was therefore based on an assumed structural resemblance to the rickardite-type phase. In this structure type, the atomic arrangement is assumed to be as follows:

$$\begin{array}{ll} 2 \text{ Me at } 0 \ 0 \ 0, & \frac{1}{2} \ \frac{1}{2} \ 0, \\ 2 \text{ Se at } 0 \ \frac{1}{2} \ z_1, & \frac{1}{2} \ 0 \ z_1, \\ x \text{ Me at } 0 \ \frac{1}{2} \ z_2, & \frac{1}{2} \ 0 \ z_2. \end{array}$$

The z_1 - and z_2 -parameters are of the order of 0.25—0.30 and 0.7, respectively, in this structure.

* Taken in part from an earlier investigation.²

Of the possible space groups for $(\text{Fe,Ni})_{22}\text{Se}_{16}$, $I\bar{4}m2$ appears to be the most fitting for the description of a structure which is comparable with that of the rickardite type, and the present determination is therefore based on this space group. The refinement of parameters was carried out by successive three dimensional Fourier syntheses, and by least squares refinements using programs written by Dahl *et al.*⁵ The full matrix least squares program was carried out using the weighting scheme No. 1:

$$\begin{aligned} \text{for } F_o \leq \text{FB}, & \quad w = A1(F_o)^{B1} \\ \text{for } F_o > \text{FB}, & \quad w = A2(F_o)^{B2} \\ \text{by taking } A1 = 1.0, & \quad A2 = 14.5, B1 = 0, B2 = -0.5 \text{ and FB} = 211. \end{aligned}$$

Only minor displacements in the Se positions occurred during the course of the work, whereas some of the metal positions were radically changed from

Table 1. Positional parameters (x,y,z), temperature factors (B), and estimated standard deviations (σ) for $(\text{Fe}_x\text{Ni}_{1-x})_{22}\text{Se}_{16}$.

	x	$\sigma(x)$	y	$\sigma(y)$	z	$\sigma(z)$	B	$\sigma(B)$
Me_I in 8(g)	0.2751	0.0008	0.2751	0.0008	0	—	0.62	0.11
Me_{II} in 8(i)	0.2784	0.0012	0	—	0.3256	0.0007	0.92	0.13
Me_{III} in 4(f)	0	—	0.5	—	0.0240	0.0011	0.90	0.18
Me_{IV} in 2(c)	0	—	0.5	—	0.25	—	0.43	0.22
Se_I in 8 (i)	0.2418	0.0008	0	—	0.1154	0.0005	0.55	0.08
Se_{II} in 8 (i)	0	—	0.2211	0.0009	0.3494	0.0004	0.43	0.08

Table 2. Observed and calculated structure factors of $(\text{Fe}_x\text{Ni}_{1-x})_{22}\text{Se}_{16}$. The columns contain $h, l, |F_o|$, and $|F_c|$, respectively.

k = 0	5	1	129	134	2	9	137	136	3	1	191	178	4	3	114	109	6	10	139	115		
h = 1	5	5	207	222	2	11	137	130	3	3	166	166	4	5	76	62	7	1	96	87		
0 2	120	83	5	7	147	158	2	13	129	130	3	5	60	52	4	7	121	108	7	3	104	89
0 4	160	148	5	9	125	125	2	15	116	105	3	9	190	197	4	9	170	171	7	7	127	115
0 6	381	394	5	13	160	177	3	0	157	154	3	11	90	84	4	11	111	98	7	9	147	131
0 8	283	276	5	15	89	72	3	2	133	141	3	13	148	134	4	15	91	104	7	11	90	101
0 12	130	116	6	0	93	87	3	4	111	114	4	2	226	227	5	0	112	105	8	0	213	228
0 18	95	81	6	2	179	193	3	8	106	105	4	4	61	52	5	2	102	105	8	4	80	71
1 1	85	70	6	6	165	188	4	3	77	68	4	6	251	256	5	4	162	161	8	8	117	98
1 3	116	121	6	10	142	150	4	5	139	153	4	10	172	169	5	6	68	66	9	5	126	120
1 5	202	209	7	1	103	105	4	7	79	74	4	12	77	67	5	8	105	104				
1 7	104	108	7	3	135	143	4	9	141	149	4	14	85	78	6	1	115	118	k = 5			
1 9	198	209	7	7	126	130	4	13	95	99	5	1	97	89	6	3	70	55	h = 1			
1 13	121	129	7	9	149	180	4	15	76	77	5	5	211	225	6	9	121	113	5	4	98	100
1 15	103	106	7	11	114	120	5	2	87	92	5	7	122	116	6	13	103	86	5	8	105	90
2 2	342	322	7	15	138	126	5	4	140	139	5	9	42	50	7	0	100	90	5	14	112	105
2 4	100	96	8	0	273	293	5	8	59	60	5	11	161	173	7	10	97	90	6	5	138	127
2 6	327	329	8	4	91	90	6	1	45	59	5	13	84	96	8	7	85	78	6	7	79	77
2 8	75	64	8	6	80	78	6	5	99	100	5	15	108	111	9	2	130	131	6	11	119	116
2 10	220	229	8	8	111	114	6	9	75	77	6	0	162	172	9	4	122	114	7	0	114	104
2 12	108	93	9	1	79	87	6	11	93	82	6	2	126	132					7	2	159	171
2 14	110	97	9	5	138	146	7	0	121	117	6	4	305	335	k = 4				7	4	138	121
2 18	79	96	9	7	103	119	7	2	123	137	7	1	131	141	h = 1				7	8	129	112
3 1	89	88	9	9	91	71	7	4	89	81	7	3	141	138	4	0	491	509	8	5	92	102
3 3	189	193	9	13	141	144	7	8	95	109	7	9	159	174	h = 1				8	5	127	129
3 5	80	77	10	0	123	130	8	5	104	85	8	2	127	129	4	6	177	167	k = 6			
3 7	111	121	10	2	87	95	8	6	140	144	8	6	140	144	4	8	180	173	h = 1			
3 9	237	241	10	6	108	108	k = 2				9	5	143	145	5	1	97	88	6	0	124	104
3 11	116	120					h = 1				10	4	115	130	5	5	169	173	6	4	207	206
3 15	127	136	k = 1				2	0	320	289					5	7	120	108	7	1	109	105
4 0	709	705	h = 1				2	2	336	315	k = 3				5	9	119	94	7	9	114	106
4 2	76	47	1	4	126	141	2	4	614	611	h = 1				5	13	149	140	8	2	106	88
4 4	120	126	2	1	114	107	2	6	78	62	3	0	157	128	6	0	60	54	8	6	105	86
4 6	239	255	2	3	95	97	2	8	70	58	3	8	108	98	6	2	166	155				
4 8	207	214	2	5	182	194	2	10	128	117	3	10	100	88	6	6	149	151				
4 12	89	75	2	7	80	78	2	12	81	71	4	1	86	73								

Table 3. Interatomic distances in $(\text{Fe}_x\text{Ni}_{1-x})_{22}\text{Se}_{16}$. The estimated standard deviations are in the range 0.006 to 0.013 Å.

From Me I (0.275 0.275 0)				From Se I (0.242 0 0.115)				
- Se I	(0.242	0	0.115)	2.39 Å	- Se I	(-0.242	0 0.115)	3.49 Å
"	(0	0.242	-0.115)	" "	"	(0	0.242 -0.115)	3.60 "
- Se II	(0.279	0.5	-0.151)	2.36 "	"	(0	-0.242 -0.115)	" "
"	(0.5	0.279	-0.151)	" "	"	(0.758	0 0.115)	3.72 "
- Me II	(0.5	0.222	-0.174)	2.59 "	"	(0.5	0.258 0.385)	4.04 "
"	(0.222	0.5	-0.174)	" "	"	(0.5	-0.258 0.385)	" "
- Me III	(0	0.5	0.024)	2.58 "	- Se II	(0	0.221 0.349)	3.56 "
"	(0.5	0	-0.024)	" "	"	(0	-0.221 0.349)	" "
From Me II (0.278 0 0.326)				From Se II (0 0.221 0.349)				
- Se I	(0.242	0	0.115)	2.41 Å	- Se II	(0	-0.221 0.349)	3.19 Å
"	(0.5	0.258	0.385)	2.54 "	"	(0.279	0.5 0.151)	3.63 "
"	(0.5	-0.258	0.385)	" "	"	(-0.279	0.5 0.151)	" "
- Se II	(0	0.221	0.349)	2.58 "	"	(0	0.779 0.349)	4.02 "
"	(0	-0.221	0.349)	" "	"	(-0.221	0 0.651)	4.11 "
- Me I	(0.225	0.225	0.5)	2.59 "	- Se I	(0.242	0 0.115)	3.56 "
"	(0.225	-0.225	0.5)	" "	"	(-0.242	0 0.115)	" "
- Me II	(0.5	0.222	0.174)	2.84 "	"	(0.5	0.258 0.385)	3.64 "
"	(0.5	-0.222	0.174)	" "	"	(-0.5	0.258 0.385)	" "
- Me III	(0.5	0	0.524)	2.77 "	"	(0.258	0.5 0.615)	4.09 "
From Me III (0 0.5 0.024)				From Se I (0.242 0 0.115)				
- Se I	(0	0.242	-0.115)	2.45 Å	"	(-0.242	0 0.115)	" "
"	(0	0.758	-0.115)	" "	"	(0.5	0.258 0.385)	3.64 "
- Se II	(0.279	0.5	0.151)	2.47 "	"	(-0.5	0.258 0.385)	" "
"	(-0.279	0.5	0.151)	" "	"	(0.258	0.5 0.615)	4.09 "
- Me I	(0.275	0.275	0)	2.58 "	- Me I	(0.225	0.225 0.5)	2.36 "
"	(0.275	0.725	0)	" "	"	(-0.225	0.225 0.5)	" "
"	(-0.275	0.275	0)	" "	- Me II	(0.278	0 0.326)	2.58 "
"	(-0.275	0.725	0)	" "	"	(-0.278	0 0.326)	" "
- Me II	(0.222	0.5	-0.174)	2.77 "	- Me III	(0	0.5 0.476)	2.47 "
"	(-0.222	0.5	-0.174)	" "	- Me IV	(0	0.5 0.25)	2.31 "
- Me IV	(0	0.5	0.25)	2.58 "				
From Me IV (0 0.5 0.25)								
- Se II	(0	0.221	0.349)	2.31 Å				
"	(0	0.779	0.349)	" "				
"	(0.279	0.5	0.151)	" "				
"	(-0.279	0.5	0.151)	" "				
- Me III	(0	0.5	0.024)	2.58 "				
"	(0	0.5	0.476)	" "				

those of the rickardite-type structure. Refinements were performed assuming anisotropic temperature factors, and attempts were made to distinguish between iron and nickel atoms. The quality of the intensity material appears to be insufficient for such determinations, however. The final refinements were therefore executed assuming isotropic temperature factors, the atomic form factors for the metal atoms being taken as the weighted mean for Fe and Ni. Refinements proceeded until no parameter shifts occurred, resulting in a reliability factor $(\sum||F_o| - |F_c|| / \sum|F_o|)$ of 0.069. The positional parameters and temperature factors of the atoms, together with estimated standard deviations, are to be found in Table 1, while the values of the observed and calculated structure factors are given in Table 2.

A perspective view of an eighth part of the unit cell ($a/2, b/2, c/2$) is shown in Fig. 1. The unit cell of the rickardite-type structure is shown in the same figure, for comparison.

Interatomic distances in $(\text{Fe}, \text{Ni})_{22}\text{Se}_{16}$ are presented in Table 3, where the shortest Me—Me distances are seen to have the substantially equal values of 2.58 and 2.59 Å. The shortest Me—Se distances are 2.31 Å, and the shortest

Se—Se distances are 3.19 Å. The metal atoms Me_{I} , Me_{III} , and Me_{IV} are coordinated with four Se atoms in a distorted tetrahedral arrangement, and Me_{II} by five Se atoms in a distorted quadrangular prism. In addition to the Se atoms, the metal atoms are surrounded by a varying number of metal atoms, and the Me—Se distances are clearly dependent on the coordination

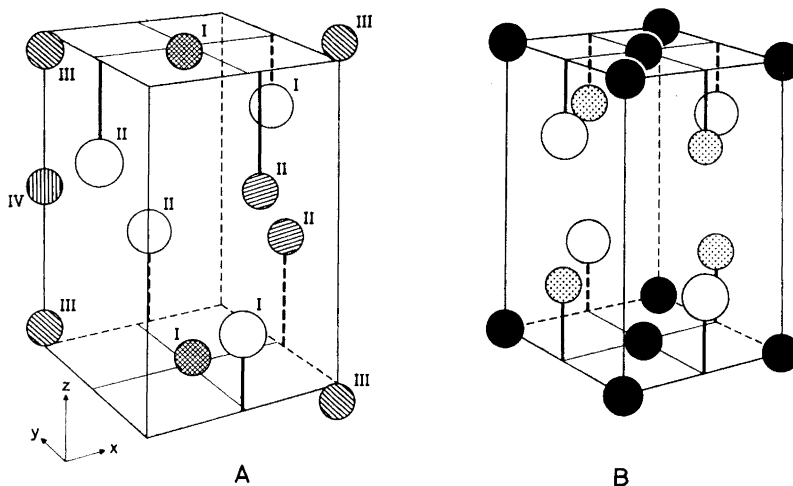


Fig. 1. A: An eighth part of the unit cell of $(\text{Fe}_x\text{Ni}_{1-x})_{22}\text{Se}_{16}$: $a/2$, $b/2$, $c/2$ Open circles represent the selenium positions, the others being those of the metal atoms. B: A unit cell of the rickardite type structure. The dotted circles represent partly occupied metal positions.

number. Thus the shortest Me—Se distances (2.31 Å) are those from the Me_{IV} atoms which have only two metal atoms in addition to the four Se atoms as nearest neighbours. The corresponding distances from Me_{I} , which is coordinated with four metal and four selenium, are 2.36 (2) and 2.39 Å (2). The Me—Se distances are even greater when measured from the positions Me_{II} and Me_{III} , in accordance with the larger coordination number.

The environments of the metal atoms in the present structure are similar to those found in the metal-rich binary phases Fe_{1+x}Se and Ni_3Se_2 . In the former, having the B10—C38 type structure, the Fe atoms are tetrahedrally coordinated with four Se atoms at 2.37 Å, and four Fe atoms at 2.66 Å^{6,7}. In the rhombohedral phase Ni_3Se_2 , the Ni atoms are tetrahedrally surrounded by four Se atoms at 2.36 Å, and by four Ni atoms at a distance of 2.57 Å.⁸ The similarity in the Me—Se distances of the latter two structures indicates that one cannot expect to distinguish between Fe and Ni atoms on the basis of interatomic distances and coordination numbers in $(\text{Fe},\text{Ni})_{22}\text{Se}_{16}$. The fact that Stevels *et al.*³ have obtained this phase in the binary system Ni—Se (as a metastable compound), may indicate that the Fe atoms are not necessarily associated with fixed positions, but possibly distributed over all of the metal positions.

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